

# Critical energy for photopolymerization of ceramic suspensions in acrylate monomers

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## Abstract

The critical energy dose  $E_c$  required for photopolymerization of silica and alumina suspensions in acrylate monomers was measured for a large series of suspensions, and contrasted with a model that based on the assumption that  $E_c$  is the dose necessary to exhaust the polymerization inhibitors. The major predictions of the model were in good agreement with the data:  $E_c$  is inversely proportional to photoinitiator concentration, directly proportional to inert dye concentration, and directly proportional to the concentration of inhibitors.

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## 1. Introduction

Photopolymerizable ceramic suspensions have been widely used in many applications such as dental restoratives<sup>1</sup> or complex 3D ceramic objects built via ceramic stereolithography.<sup>2,3</sup>

Ceramic suspensions of powders in monomer solutions containing photoinitiators are photopolymerized by exposing the surface to actinic light, such as ultraviolet, at a dose  $E$  (J/m<sup>2</sup>) that suffices to polymerize the suspension to a certain depth of cure  $C_d$ .<sup>4</sup> The cure depth is related to the logarithm of the energy dose<sup>5</sup> by  $C_d = D_p \ln(E/E_c)$  where the photobehavior of the suspension is characterized by a critical energy  $E_c$  and a sensitivity  $D_p$ . The photobehavior parameters will depend upon the compositional variables in a ceramic suspension, which includes the amount and type of ceramic powder, the monomer solution, and the concentration and type of the photoactive agents, such as the photoinitiators, inert dyes, and polymerization inhibitors. In another paper,<sup>6</sup> we derived simple predictive models which relate the critical energy and resin sensitivity to the suspension composition, and presented a limited amount of data. In this paper we consider the critical energy model in more detail,

contrast its predictions with experiment and evaluate the fundamental parameters and compositional variables that determine the critical energy. A companion paper<sup>7</sup> reports details on the sensitivity  $D_p$ .

Free-radical photopolymerization occurs when radicals created by the photoinitiator are present in sufficient quantity to initiate the polymerization reaction of the monomers.<sup>8</sup> However, monomers must be stable against small quantities of free radicals that might be inadvertently generated. Thus the monomer solutions contain inhibitors which are added to prevent an undesired reaction. Oxygen absorbed from the air can also behave as an inhibitor. The model for  $E_c$  was based on the presumption that for UV doses smaller than  $E_c$ , the free radicals created when a photoinitiator molecule absorbs a photon are consumed by inhibitors so polymerization does not occur. It also considers absorption of photons by inert dyes. The hypothesis of this model for critical energy is that  $E_c$  is given by the dose of UV photons that are not absorbed by inert dyes and produce a population of free radicals large enough to exhaust the population of inhibitors in the monomer solution to provide surplus free radicals to begin the polymerization reaction. The critical energy then depends upon the number and effectiveness of inhibitors that scavenge free radicals, the concentration and absorption coefficients of inert dyes that absorb photons, and the concentration and absorption coefficients of photoinitiators that create free radicals when they absorb photons. While it is the monomer suspension medium that is photo-chemically active, the pres-

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### Nomenclature

$C_d$	cure depth
$c_P$	molar concentration of photoinitiator
$c_D$	molar concentration of dye
$D_p$	penetration depth
$E_c$	critical energy
$\varepsilon_D$	molar extinction coefficient of the dye
$\varepsilon_P$	molar extinction coefficient of the photoinitiator is $\varepsilon_P$
$\Phi$	ceramic volume fraction
$\gamma_D$	the number of radicals that did not get generated because the photon was absorbed by an inert dye
$\gamma_Q$	potency of the inhibitor, or the number of radicals removed per inhibitor
$\gamma_O$	potency of the oxygen inhibitor, or the number of radicals removed per oxygen inhibitor
$h\nu$	photon energy
$l_{Sc}$	scattering length
$O$	concentration of oxygen inhibitor
$\Omega$	quantum yield
$Q$	concentration of quinone-type inhibitor

ence of ceramic powders plays a passive role by diluting the active monomer, and an active role by limiting photon penetration depth by scattering. These effects were considered in an “inhibitor exhaustion model” for the critical energy parameter  $E_c$  which was derived in the previous paper.<sup>6</sup> The inhibitor exhaustion model suggests that the critical energy should depend on the compositional variables of the suspension, including the ceramic volume fraction  $\Phi$ , the concentration of the photoinitiator ( $c_P$  in mol/L), the inert dye concentration ( $c_D$  in mol/L) and the concentration of quinone-type inhibitor ( $Q$ ) and oxygen inhibitor ( $O$ ), as:

$$E_c = (1 - \Phi) \frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O + \gamma_D c_D) \frac{1}{\varepsilon_P c_P} + \frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O + \gamma_D c_D) \times \left[ \frac{(1/l_{Sc}) + (1 - \Phi)(\varepsilon_D c_D)}{(1 - \Phi)\varepsilon_P^2} \right] \frac{1}{c_P^2} \quad (1)$$

The other terms in Eq. (1) are as follows:  $\Omega$  is the quantum yield, or number of free radicals produced by absorption of a photon of energy  $h\nu$ ; the molar extinction coefficient of the photoinitiator is  $\varepsilon_P$ ;  $\varepsilon_D$  is the molar extinction coefficient of the dye;  $\gamma_Q$  is the potency of the inhibitor, or the number of radicals removed per inhibitor; and  $\gamma_O$  is the potency of the oxygen inhibitor, or the number of radicals removed per oxygen inhibitor; light scattering from the particles is characterized by the scattering length  $l_{Sc}$ .

Eq. (1) is written as a sum of terms linear in  $1/c_P$  as the first term and terms quadratic in  $1/c_P$  as the second term. It can be simplified if it happens that the quadratic term is much smaller than the linear term. If that is the case the quadratic term can

be neglected, and the critical energy can be approximated in a linearized form:

$$E_c = (1 - \Phi) \frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O + \gamma_D c_D) \frac{1}{\varepsilon_P c_P} \quad (2)$$

The linearized form of the inhibitor exhaustion model makes several predictions that can be easily tested. It predicts that the critical energy should depend on the inverse of photoinitiator concentration as  $1/c_P$ , so that higher PI concentration should have smaller critical energy. A plot of  $E_c$  vs.  $1/c_P$  should be straight line with zero intercept whose slope ( $K_3$  in J-mol/m<sup>2</sup>L) reports the values of the terms in Eq. (3):

$$\frac{dE_c}{d(1/c_P)} = K_3 = (1 - \Phi) \frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O + \gamma_D c_D) \frac{1}{\varepsilon_P} \quad (3)$$

Eq. (2) also predicts that a plot of  $E_c$  vs. inert dye concentration should be linear. The intercept ( $I_4$ ) should be given by Eq. (4a) and the slope ( $K_4$  in J-L/m<sup>2</sup> mol) by Eq. (4b):

$$E_{c \text{ for } (c_D=0)} = I_4 = (1 - \Phi) \frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O) \frac{1}{\varepsilon_P c_P} \quad (4a)$$

$$\frac{dE_c}{dc_D} = K_4 = (1 - \Phi) \frac{h\nu}{\Omega} \frac{\gamma_D}{\varepsilon_P c_P} \quad (4b)$$

The critical energy should increase linearly with additions of inhibitors. Since there is an unknown amount of pre-existing inhibitor ( $Q_{native}$ ), the total inhibitor concentration has both the native and added inhibitor, so we expect a plot of  $E_c$  vs.  $Q_{added}$  to have a slope ( $K_5$ ) and intercept ( $I_5$ ) given by Eqs. (5a) and (5b):

$$\frac{dE_c}{dQ_{added}} = K_5 = (1 - \Phi) \frac{h\nu}{\Omega} \frac{\gamma_Q Q_{added}}{\varepsilon_P c_P} \quad (5a)$$

$$E_{c \text{ for } (Q_{added}=0)} = I_5 = (1 - \Phi) \frac{h\nu}{\Omega} (\gamma_Q Q_{native} Q_{native} + \gamma_O O + \gamma_D c_D) \frac{1}{\varepsilon_P c_P} \quad (5b)$$

The linearized inhibitor exhaustion model also predicts that the only effect of the ceramic comes from the volume fraction term, so  $E_c$  vs. the volume fraction ceramic should have a slope and intercept from Eqs. (6a) and (6b):

$$\frac{dE_c}{d\Phi} = K_6 = -\frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O + \gamma_D c_D) \frac{1}{\varepsilon_P c_P} \quad (6a)$$

$$E_{c, \Phi=0} = I_6 = \frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O + \gamma_D c_D) \frac{1}{\varepsilon_P c_P} \quad (6b)$$

so the slope and intercept should be *numerically equal*. Another prediction of the linearized model is that *all ceramics should behave similarly*. This is a consequence of neglecting the quadratic terms in Eq. (1), which contained the scattering length, which depends on the volume fraction, particle size and refractive index contrast. Different ceramic powders can have quite different values for the scattering length,<sup>9,10</sup> but since this term is not included in the simple linearized model, we expect strong

scattering powders (like alumina) to be have similarly as weakly scattering powders (like silica).

In this paper, we will test the predictions of the simple linearized model with experimental measurements of  $E_c$  varying the concentrations of photoinitiators, dyes, and inhibitors. Note that the terms in Eqs. (3)–(6) for the slopes and intercept appear repeatedly, so each independent series of experiments provides another value for these collections of terms. We will seek a self-consistent set of terms to enable us to predict the properties of unknown suspensions from the experimentally determined  $E_c$  values of tested suspensions. Some of the parameters can be independently measured, such as the molar extinction coefficients for the dye and the photoinitiator, which can be determined with spectrophotometry. We will compare a limited number of ceramic volume fractions with silicon dioxide suspensions, and present a small amount of data on aluminum oxide suspensions, and discuss the successes and limitations of the linearized inhibitor exhaustion model.

## 2. Experimental procedure

### 2.1. Method for determining critical energy

The critical energy was determined measuring the cure depth as a function of energy dose. The surfaces of photosuspensions were exposed to six different energy doses to make thin coupons of solid green ceramic. The energy doses ranges were chosen mostly with respect to the concentration of PI and dye. For example 60 vol%  $\text{SiO}_2$  suspension with ketone PI and no dye was polymerized using the energy range 60–400  $\text{mJ}/\text{cm}^2$ , while for example 55 vol%  $\text{Al}_2\text{O}_3$  suspension with 0.0252 mol/L (wrt monomer) ketone PI and 0.0026 mol/L (wrt monomer) triazole dye required much higher energy doses  $\sim 500$ – $3300$   $\text{mJ}/\text{cm}^2$ . The coupons were lifted from the liquid photosuspensions and rinsed with isopropanol to remove uncured suspension. The cured thickness  $C_d$  was measured with a micrometer on 5 different spots of the sample. For the  $E_c$  calculation, the minimum value was used. This approach minimizes errors that can arise due to improper cleaning technique and warping of the sample due to polymerization shrinkage. Each composition was measured  $\sim 2$ – $3$  times. The cure depths were on the order of mm in case of silica suspension with no dye or tens to hundreds of microns in case of  $\text{Al}_2\text{O}_3$  suspension with dye. Coupons with smaller cure depths were more difficult to handle and had more variation.

The cure depths were plotted against the logarithm of the energy dose. A linear regression line was fit through the data set to infer the sensitivity  $D_p$  from the regression slope. The regression line was extrapolated to the  $C_d = 0$  to infer  $E_c$  from the intercept. As it involves extrapolation on a semi-log plot with six points, the values of  $E_c$  determined were quite sensitive to the details of the regression line. The standard errors for the slope  $D_p$  and intercept  $\ln(E_c)$  were estimated from the residual sum of squares, using a statistical analysis package (OriginPro 8, OriginLab Corporation, North Hampton, MA, USA) to estimate the average  $\ln(E_{c\text{average}})$  and the residual sum of squares to

estimate of the standard deviation of the  $\ln(E_c)$  as  $\delta \ln(E_c)$ . When these are expressed as  $E_c$  rather than  $\ln(E_c)$ , the error limits are quite asymmetric, so are reported as  $E_{c\text{average}}$  with limits  $E_{c\text{min}}$  and  $E_{c\text{max}}$ .

Most photopolymerization experiments were performed by laser scanning in a stereolithography apparatus (SLA-250, 3D systems, Inc., Valencia, CA, USA) with a solid state laser (Xcyte, JDSU, Milpitas, CA, USA), that had a quasi-continuous wave emitting at 355 nm, output power 30 mW, and a beam diameter of 125  $\mu\text{m}$ . The intensity of the laser beam was measured before each experiment using the sensor built into the SLA-50, which had been calibrated with a thermal laser power meter (Orion-TH, Ophir Optronics, Jerusalem, Israel). A standard SLA test protocol was used to vary the dose by changing the scanning rate and hatch spacing. Other experiments were performed using a UV curing apparatus (Conveyorized UV Curing System, Hanovia, Newark, NJ, USA) equipped with 300 W mercury vapor lamps. Hanovia lamps have strong lines at 305, 315, 365, 405 and 415 nm. The UV exposure dose was calibrated using a radiometer (UV-integrator PC-2008, Hanovia).

### 2.2. Materials

#### 2.2.1. Photoactive agents

Photopolymerizable suspensions in this study are based on photocurable monomers 1,6-hexanediol diacrylate (SR238, Sartomer, USA) and ethoxylated pentaerythritol tetraacrylate (SR494, Sartomer, USA). Both are fast curing acrylate monomers and were used as-received. The diacrylate is a bifunctional monomer of low viscosity (9 mPa s), density 1.02  $\text{g}/\text{cm}^3$ , molecular weight 226 g/mol and provides hardness when polymerized. The tetraacrylate has viscosity 108 mPa s, density 1.128  $\text{g}/\text{cm}^3$  and molecular weight 528 g/mol.

The photoactive agents included free-radical producing photoinitiators and inert absorbing dyes. The photoinitiator used for most experiments was a non-photobleaching ketone (Irgacure 184, Ciba, USA). Irgacure 184 is 1-hydroxycyclohexyl-phenyl-ketone and is a solid white powder with density 1.1–1.2  $\text{g}/\text{cm}^3$ . Some experiments were conducted with a photobleaching phosphine oxide photoinitiator (Irgacure 819, Ciba, USA). Irgacure 819 is phosphine oxide, phenylbis(2,4,6-trimethyl benzoyl). It is a solid light yellow powder with density 1.205  $\text{g}/\text{cm}^3$ . The inert UV dye was a commercial UV absorber (Tinuvin 171, Ciba, USA). Tinuvin 171 belongs to the hydroxyphenylbenzotriazole class and its main component is phenol, 2-(2H-benzotriazol-2-yl)-6-dodecyl-4-methyl-, branched and linear. It is a yellow liquid with pH 5.4 and density 1.003  $\text{g}/\text{cm}^3$ . A visible dye, Thermoplast yellow 104 (BASF, USA) was used as a “blue light absorber”. It is 3H-pyrazol-3-one,4-[(1,5-dihydro-3-methyl-5-oxo-1-phenyl-4H-pyrazol-4-ylidene)methyl]-2,4-dihydro-5-methyl-2-phenyl. It is a yellow powder and exhibits good solubility in these acrylate monomers. The absorbance spectra for all the photoactives are reported in a companion paper,<sup>7</sup> where we also report values of the molar absorption coefficients obtained by convoluting the absorption spectra with the emission lines of the laser (355 nm) and the emission spectrum of the mercury vapor lamps.

Table 1  
Properties of used photoinitiators, dyes, radical scavengers and inhibitors.

	Commercial name	Density (g/cm <sup>3</sup> )	Molecular weight (g/mol)
Photoinitiator	Irgacure 184	1.1–1.2	204.3
Photoinitiator	Irgacure 819	1.205	418.5
UV light absorber	Tinuvin 171	1.003	~395
Visible light absorber	Thermoplast yellow 104	1.3	358.5
HALS stabilizer	Tinuvin 123	0.97	737
HALS stabilizer	Irgastab UV 10	1.12	510.7
Inhibitor	Methoxy hydroquinone	–	140.14

Two types of inhibitors were used to as radical scavengers. Two were commercial hindered amine light stabilizers (HALSs) (Tinuvins 123 and Irgastab UV 10, Ciba, USA). Tinuvins 123 was a clear, slightly yellow liquid exhibiting viscosity  $\sim 3$  Pa s at 20 °C. Irgastab UV 10 is an orange crystalline powder and has density 1.12 g/cm<sup>3</sup>. The other inhibitor was methoxy hydroquinone (2,5-dihydroxyanisole, Sigma–Aldrich, USA) and was used as inhibitor. Table 1 summarizes the properties of the photoinitiators, dyes, radical scavengers and inhibitors.

### 2.2.2. Ceramic suspensions

The ceramic powder was silicon oxide, 99.8%, metal basis (Alfa Aesar, USA). This is a fused silica with mean diameter 7.1  $\mu$ m, a specific area of 5 m<sup>2</sup>/g and density 2.2 g/cm<sup>3</sup> (all from the manufacturer's specifications) and was used as-received without further purification. Alumina powder (A16-SG, Alcoa, USA) with particles mean diameter  $d_{50} = 0.4$   $\mu$ m, specific surface area 8.6 m<sup>2</sup>/g and density 3.92 g/cm<sup>3</sup> (from the manufacturer's specifications) was used as-received without further purification. The colloidal dispersant for these powder in the acrylate monomers was a commercial alkoxyated ammonium phosphate dispersant (Variquat CC-59, Evonik Degussa GmbH). The dispersant loading was 2.083 wt% with respect to the dry powder mass.

The ceramic suspensions were prepared by the ball milling. First, alumina milling media (1/4 in. diameter) were added into an opaque polyethylene bottle (size 250 mL) in the amount of 1/15 of the total volume of the bottle. The diacrylate and tetraacrylate monomers were added at the volume ratio 9/1 along with the dispersant. The mixture was ball milled for  $\sim 15$  min at 30 rpm to produce a well blended system. Ceramic powder was added incrementally – one quarter at a time and after each powder addition the suspension was ball milled for at least 3 h. When desired vol% solid loading was achieved, the suspensions were ball milled additional 24 h. Photoinitiator was added last and suspensions were ball milled for other 6 h. In case of suspensions containing UV absorber, HALS stabilizer or inhibitor, these were added last and ball milled for one additional day. The total volume of the suspension was calculated to be one half of the total volume of the bottle.

## 3. Results

To understand critical energy and accuracy of the measurements, raw cure depth measurements data have to be explained first. Shown in Fig. 1 is the cure depth  $C_d$  vs. energy dose

logarithm for 60 vol% SiO<sub>2</sub> suspension with ketone PI (0.2016 mol/L wrt monomer) and no dye. The graph contains 3 sets of cure depth-energy dose data, when each of the set includes 6 measurements at different energy dose. As can be seen from the graph, polymerization using the same energy dose results in variations in the  $C_d$  value. The variations in  $C_d$  are caused by several factors such as laser intensity fluctuations within 5–10%, accuracy of samples cleaning, accuracy of the micrometer and thickness measurement error.

According to Jacob's equation, the data can be fitted with linear regression line for  $C_d$  vs.  $\ln(E)$ , where the slope represents the sensitivity  $D_p$  and the  $x$ -intercept represents the critical energy  $E_c$ . Thus, average  $D_p$  and average  $E_c$  are obtained. Using a standard analysis of the linear regression with 95% confidence of interval, the variations in the penetration dept  $\delta D_p$  and critical energy  $\delta E_c$  can be determined. The variations in the  $D_p$  are within  $\pm 5\%$ . On the other hand,  $E_c$  is "measured" by extrapolating the linear curve on a semi-log plot. Therefore the experimental determination of the critical energy by itself is not particularly accurate resulting in significantly larger errors as can be seen in Fig. 1. Moreover, the error bars are asymmetric because of the logarithm and they become more apparent when higher critical energy doses are required. Error bars were calculated on each individual composition.

The critical energy  $E_c$  was determined on a large series of suspensions with varying ceramic volume fraction, photoinitiator and dye concentration in order to test the proposed model. The linearized inhibitor exhaustion model predicts that  $E_c$  plotted against  $1/C_p$  should be a straight line with zero intercept. We expect  $E_c$  to decrease with increasing photoinitiator concentration. The slope should be steeper with lower ceramic volume, larger inhibitor or dye concentration. To test these predictions, the critical energy of a series of suspensions was plotted against

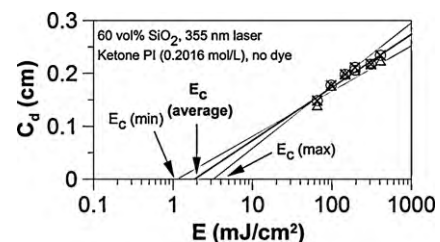


Fig. 1. Cure depths vs. energy dose for 60 vol% SiO<sub>2</sub> suspension with ketone PI (0.2016 mol/L) and no dye. Cure depth measurements were performed using 355 nm laser. The data was fitted with  $y = K \cdot \ln x + q$ , where the slope represents sensitivity  $D_p$  and the  $x$ -intercept the critical energy  $E_c$ . The average energy is 1.89 mJ/cm<sup>2</sup>,  $E_{min} = 1.19$  mJ/cm<sup>2</sup> and  $E_{max} = 3.28$  mJ/cm<sup>2</sup>.

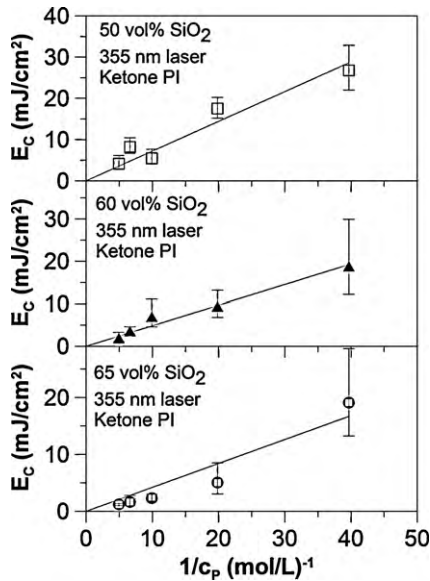


Fig. 2. Critical energy vs.  $1/c_p$  for 50, 60 and 65 vol% SiO<sub>2</sub> suspensions with ketone PI and no dye.

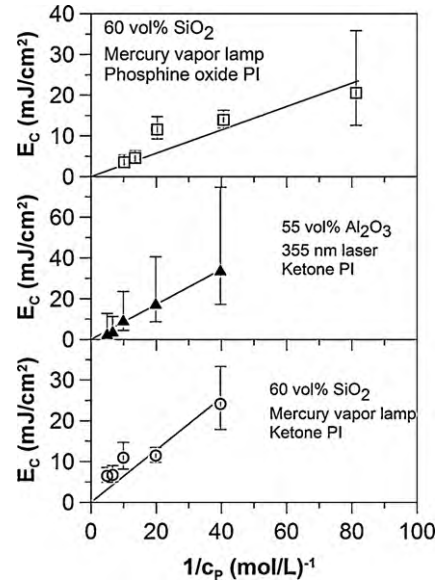


Fig. 3. Critical energy vs.  $1/c_p$  for 60 vol% SiO<sub>2</sub> suspensions with the ketone and phosphine oxide PI and 55 vol% Al<sub>2</sub>O<sub>3</sub> suspension with the ketone PI. Suspensions were prepared with no dye.

$1/c_p$  and fitted with linear regression with intercept forced to have zero intercept.

Shown in Fig. 2 are a series of plots of  $E_c$  vs.  $1/c_p$  for 50, 60 and 65 vol% SiO<sub>2</sub> suspensions with ketone PI polymerized using the 355 nm laser. Similarly, shown in Fig. 3 is the  $E_c$  vs.  $1/c_p$  for SiO<sub>2</sub> suspensions with the ketone PI and phosphine oxide PI polymerized using the mercury vapor lamp and Al<sub>2</sub>O<sub>3</sub> suspensions with ketone PI polymerized using the 355 nm laser. The suspensions contain no dye ( $c_D = 0$ ). In all cases, the data was satisfactorily represented by a straight line with zero intercept. It appears that the prediction that  $E_c \sim 1/c_p$  agrees with these observations. Consider now the actual slopes, which are presented in Table 2. In case of the silica suspensions with ketone PI polymerized using 355 nm laser, the slope increases with decreasing ceramic volume content as the model predicts. The slope term  $K_3/(1 - \Phi)$  should be the same for all compositions containing the same PI since the suspensions are based on the same monomer solution and contain similar concentration of the “native” inhibitors. The slope term  $K_3/(1 - \Phi)$  for suspensions with the ketone PI were roughly the same. The all in the range from 0.7 to 1.76 (mJ·mol)/(L·cm<sup>2</sup>), with an average of 1.4 (mJ·mol)/(L·cm<sup>2</sup>). Suspensions with the phosphine oxide PI had the  $K_3/(1 - \Phi)$  term of 0.89 (mJ·mol)/(L·cm<sup>2</sup>).

The addition of an inert dye increases the critical energy. The  $E_c$  vs.  $1/c_p$  curves for 60 and 50 vol% SiO<sub>2</sub> suspensions with triazole dye is shown in Figs. 4 and 5, respectively. The suspensions contain ketone PI and were polymerized using 355 nm laser. Again the critical energy values are represented well by a straight line with zero intercept. The actual slopes and correlation coefficients for individual compositions with the triazole dye are summarized in Table 3. From Eq. (3), we expect that the slope term  $K_3/(1 - \Phi)$  should increase with the concentration of inert dye. That is what is observed for the first 3 dye concentrations, but the slope term is smaller for the highest dye concentration. By excluding the portion of the energy consumed by the quinone inhibitor and oxygen inhibition determined previously ( $\sim 1.39$  (mJ·mol)/(L·cm<sup>2</sup>)), the dye term  $(h\nu/\Omega)(\gamma_D/\epsilon_P)$  was estimated to be 1530 mJ/cm<sup>2</sup>.

The simple linear model also predicts that, for a fixed concentration of photoinitiator, the critical energy dose should be a linear function of the inert dye concentration  $c_D$ . The slope and intercept of a plot of  $E_c$  vs.  $c_D$  should decrease with increasing PI concentration. Fig. 6 shows  $E_c$  vs. dye concentration for 50–65 vol% SiO<sub>2</sub> with ketone PI and triazole dye polymerized using 355 nm laser. The cure depth measurements were per-

Table 2

Actual slopes and correlation coefficients of linear regressions  $E_c$  vs.  $1/c_p$  for SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> suspensions with no dye ( $c_D = 0$ ). Intercept was set to zero.

Light source	Ceramic	PI	$\Phi$	$r^2$	$K_3$	$\frac{K_3}{1-\Phi} = \frac{h\nu}{\Omega}(\gamma_Q Q + \gamma_O O) \frac{1}{\epsilon_P} \frac{K_3}{(1-\Phi)} = \frac{h\nu}{\Omega}(\gamma_Q Q + \gamma_O O) \frac{1}{\epsilon_P}$ (mJ·mol)/(L·cm <sup>2</sup> )
355 nm laser	SiO <sub>2</sub>	Ketone	0.5	0.94	0.785 ± 0.088	1.57 ± 0.18
	SiO <sub>2</sub>		0.6	0.98	0.499 ± 0.100	1.25 ± 0.25
	SiO <sub>2</sub>		0.65	0.99	0.248 ± 0.026	0.71 ± 0.07
	Al <sub>2</sub> O <sub>3</sub>		0.55	0.95	0.794 ± 0.552	1.76 ± 1.23
Mercury vapor lamp	SiO <sub>2</sub>	Ketone	0.6	0.9	0.666 ± 0.085	1.66 ± 0.21
Average						1.39
Mercury vapor lamp	SiO <sub>2</sub>	Phosphine oxide	0.6	0.95	0.357 ± 0.047	0.89 ± 0.12

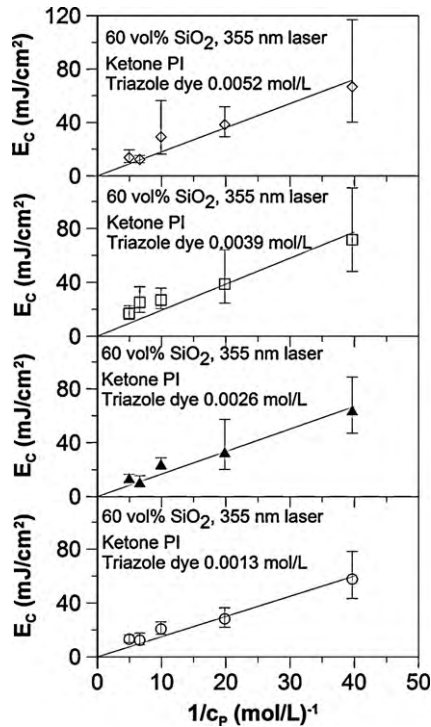


Fig. 4. Critical energy vs.  $1/c_p$  for 60 vol% SiO<sub>2</sub> suspensions with ketone PI and varying concentration of triazole dye (wrt monomer).

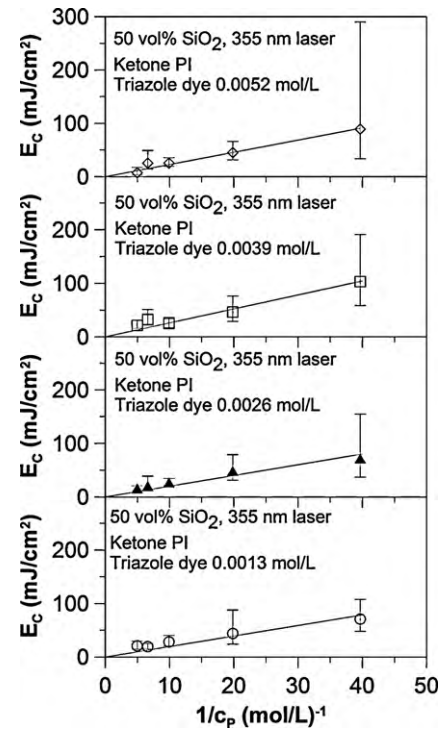


Fig. 5. Critical energy vs.  $1/c_p$  for 50 vol% SiO<sub>2</sub> suspensions with ketone PI and varying concentration of triazole dye (wrt monomer).

formed using the 355 nm laser. Fig. 7 has data for 55 vol% Al<sub>2</sub>O<sub>3</sub> suspensions with ketone PI, cure depth measurements were performed using the 355 nm laser. Fig. 7 also has data for 60 vol% SiO<sub>2</sub> with phosphine oxide PI (0.0984 mol/L wrt monomer) and blue light absorber (cure depth measurements performed using mercury vapor lamp). For clarity, the data in the graph are shown without error bars, although the linear regression lines considered the variations in  $E_c$ . Within some scatter, the critical energy increases linearly with inert dye concentration. The correlation coefficients, slopes and intercepts are shown in Table 4. Fig. 8 shows  $I_4/(1 - \Phi)$  vs.  $1/c_p$  and  $K_4/(1 - \Phi)$  vs.  $1/c_p$  for 50, 60 and 65 vol% SiO<sub>2</sub> suspensions. Both terms linearly increase with increasing PI concentration, which is in good agreement with the models prediction. Further, the intercept term  $I_4 c_p/(1 - \Phi)$  should be the same for all compositions (with the same PI), since the compositions are based on the same monomer solution and therefore contain the same amount of the native inhibitors.

Table 4 shows that for suspensions with the ketone photoinitiator this intercept term  $I_4 c_p/(1 - \Phi)$  varied from about 0.7–2.7 (mJ-mol)/(L-cm<sup>2</sup>) with an average around 1.45 (mJ-mol)/(L-cm<sup>2</sup>). The intercept term was 0.96 (mJ-mol)/(L-cm<sup>2</sup>) for suspensions with phosphine oxide PI. The dye term  $(h\nu/\Omega)(\gamma_D/\epsilon_P)$  was estimated to be 1400 mJ/cm<sup>2</sup>, which agrees with the dye term determined from Table 3.

The intercept term  $I_4 c_p/(1 - \Phi)$  from the dye concentration series of experiments (Table 4) should be numerically equal to the slope term  $K_3/(1 - \Phi)$  for suspensions with no dye from the PI concentration series of experiments (Table 2). In fact, these are in good agreement. For the ketone PI, the average value of  $I_4 c_p/(1 - \Phi)$  is 1.45 (mJ-mol)/(L-cm<sup>2</sup>), while the average value of  $K_3/(1 - \Phi)$  is 1.39 (mJ-mol)/(L-cm<sup>2</sup>). For the phosphine PI, the values are 0.96 (mJ-mol)/(L-cm<sup>2</sup>) from Table 4 and 0.89 (mJ-mol)/(L-cm<sup>2</sup>) from Table 2.

Table 3

Actual slopes and correlation coefficients of linear regressions  $E_c$  vs.  $1/c_p$  for 50 and 60 vol% SiO<sub>2</sub> suspensions with ketone PI and triazole dye.

Light source	Ceramic	$\Phi$	$c_D$ (mol/L)	$r^2$	$K_3$	$\frac{K_3}{1-\Phi} = \frac{h\nu}{\Omega}(\gamma_Q Q + \gamma_O O + \gamma_D c_D) \frac{1}{\epsilon_P} \frac{K_3}{(1-\Phi)} = \frac{h\nu}{\Omega}(\gamma_Q Q + \gamma_O O + \gamma_D c_D) \frac{1}{\epsilon_P}$ (mJ-mol)/(L-cm <sup>2</sup> )	$\frac{h\nu}{\Omega} \frac{\gamma_D}{\epsilon_P}$ (mJ/cm <sup>2</sup> )	
355 nm laser	SiO <sub>2</sub>	0.6	0.0013	0.93	1.77 ± 0.24	4.4 ± 0.61	2300 ± 300	
			0.0026	0.94	2.16 ± 0.29	5.4 ± 0.72	1550 ± 200	
			0.0039	0.92	2.57 ± 0.50	6.4 ± 1.26	1300 ± 250	
			0.0052	0.97	1.97 ± 0.35	4.9 ± 0.9	700 ± 100	
	SiO <sub>2</sub>	0.5	0.0013	0.92	2.64 ± 0.53	5.3 ± 1.0	3000 ± 600	
			0.0026	0.98	2.77 ± 0.53	5.6 ± 1.0	1600 ± 300	
			0.0039	0.89	2.93 ± 0.75	5.9 ± 1.5	1150 ± 300	
			0.0052	0.96	2.36 ± 0.68	4.7 ± 1.4	600 ± 200	
			Average					1530

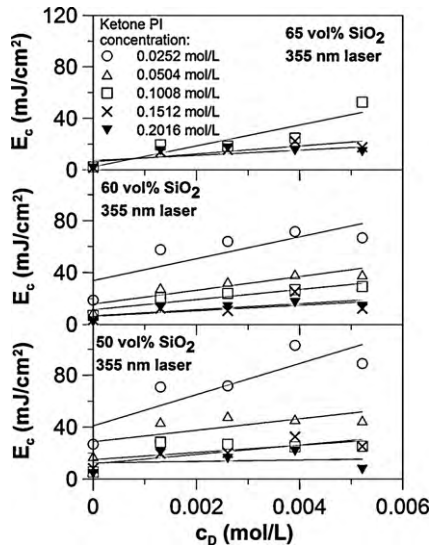


Fig. 6. Critical energy vs.  $c_D$  for 50, 60 and 65 vol% SiO<sub>2</sub> suspensions with triazole dye and varying ketone PI concentration (wrt monomer).

The linearized model further predicts that addition of inhibitors should produce a linear curve when  $E_c$  is plotted against  $Q_{added}$ . Here, we studied the effect of one inhibitor MEHQ and two HALSs that scavenge free radicals and behave similarly as inhibitors as their effect on sensitivity and critical energy. The effect of inhibitor and HALSs on critical energy is shown in Fig. 9 for 60 vol% SiO<sub>2</sub> suspensions with the inhibitor MEHQ, the solid orange HALS and the liquid yellow HALS. The compositions contain ketone PI (0.1008 mol/L wrt monomer). As can be seen from the graph, the data resulted in linear lines, which is in good agreement with the model. The solid orange HALS dramatically increases the critical energy even in small concentrations, while the liquid yellow HALS and MEHQ do not significantly affect the critical energy. Moreover,

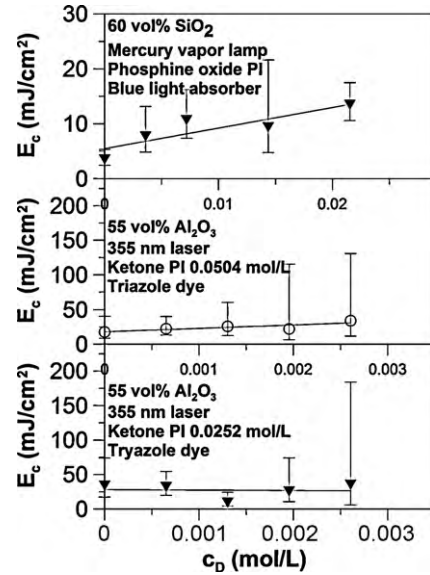


Fig. 7. Critical energy vs.  $c_D$  for 60 vol% SiO<sub>2</sub> suspension with phosphine oxide PI and blue light absorber and 55 vol% Al<sub>2</sub>O<sub>3</sub> suspension with ketone PI and triazole dye.

in case of the liquid yellow HALS the slight increase in the critical energy can be attributed to the reduction of ceramic volume content from 60 to ~56 vol% for a composition with the highest concentration of the liquid yellow HALS, since the HALSs were added after the suspensions were prepared.

Slopes, intercepts and correlation coefficients of the regression lines are summarized in Table 5. The intercept term  $I_{5CP}/(1 - \Phi)$  reports on the portion of the energy consumed by the native inhibitors and by oxygen inhibition. Its value inferred from the added-inhibitor series was determined to be 2 (mJ-mol)/(L-cm<sup>2</sup>). The model predicts that this value should be the same as  $I_{4CP}/(1 - \Phi)$  from Table 4, which is

Table 4

Actual slopes, intercepts and correlation coefficients of linear regressions  $E_c$  vs.  $c_D$  for SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> suspensions.

Light source	Ceramic	$\Phi$	PI	$c_P$ (mol/L)	$r^2$	$K_4$	$I_4$	$\frac{K_{4CP}}{(1-\Phi)} = \frac{h\nu \gamma_D}{\Omega \epsilon_P} \frac{K_{4CP}}{1-\Phi} = \frac{I_{4CP}}{1-\Phi} = \frac{h\nu}{\Omega} (\gamma_Q Q + \gamma_O O) \frac{1}{\epsilon_P}$ (mJ-mol)/(L-cm <sup>2</sup> )		
355 nm laser	SiO <sub>2</sub>	0.5	Ketone	0.0252	0.85	21,500 ± 14,600	27 ± 6	1100 ± 700		
				0.0504	0.84	6400 ± 3500	18 ± 3	650 ± 350		
				0.1008	0.71	5200 ± 1500	6 ± 2	1100 ± 300		
				0.1512	0.82	5800 ± 2600	9 ± 2	1700 ± 800		
				0.2016	0.42	3100 ± 1300	5 ± 2	1250 ± 500		
	SiO <sub>2</sub>	0.6	Ketone	0.0252	0.74	14,000 ± 6300	23 ± 10	900 ± 400		
				0.0504	0.77	6600 ± 2400	11 ± 4	800 ± 300		
				0.1008	0.83	5600 ± 1900	9 ± 4	1400 ± 500		
				0.1512	0.69	2000 ± 600	4 ± 1	700 ± 200		
				0.2016	0.74	3500 ± 500	2 ± 1	1800 ± 300		
	SiO <sub>2</sub>	0.65	Ketone	0.1008	0.87	8900 ± 2000	2 ± 1	2600 ± 600		
				0.1512	0.75	4400 ± 1300	2 ± 1	1900 ± 600		
				0.2016	0.78	4100 ± 600	1 ± 0	2400 ± 350		
				Average					1400	1.45
				Average 355 nm laser	Al <sub>2</sub> O <sub>3</sub>	0.55	Ketone	0.0252	-0.13	(-8000) ± 9000
0.0504	0.84	5700 ± 23,900	18 ± 19					600 ± 2700		
Average Mercury vapor lamp	SiO <sub>2</sub>	0.6	Phosphine oxide	0.0984	0.87	500 ± 200	4 ± 2	120 ± 50		

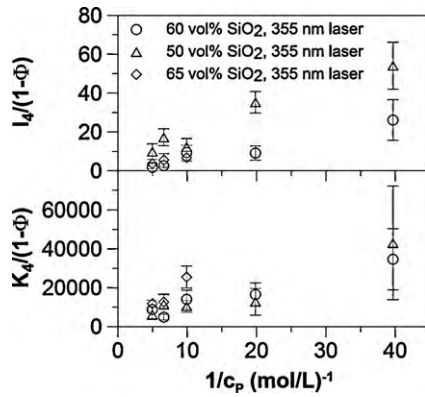


Fig. 8. Slope and intercept determined from linear regressions of  $E_c$  vs.  $c_D$  for 50, 60 and 65 vol%  $\text{SiO}_2$  suspensions.

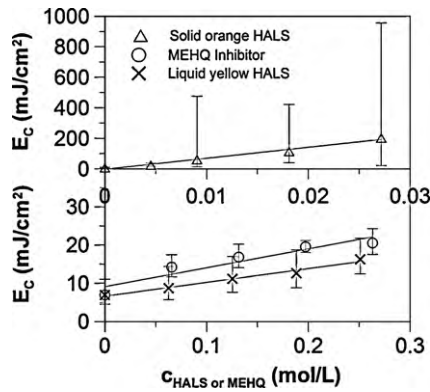


Fig. 9. Critical energy vs. concentration of inhibitor resp HALS (wrt monomer) for ~60 vol%  $\text{SiO}_2$  suspension containing the ketone photoinitiator (0.1008 mol/L wrt monomer). Cure depth measurements were performed using the 355 nm laser.

1.45 (mJ-mol)/(L-cm<sup>2</sup>) and  $K_3/(1 - \Phi)$  from Table 2, which is 1.39 (mJ-mol)/(L-cm<sup>2</sup>). These are roughly the same. The slope of Fig. 9 is the term  $K_5 c_P/(1 - \Phi)$ , which determines the portion of the energy consumed by the HALSs or the added inhibitor. The portion of the energy absorbed by the solid orange HALS, the liquid yellow HALS and inhibitor MEHQ was estimated to be  $1033 \pm 756$ ,  $12 \pm 4$  and  $9 \pm 6$  mJ/cm<sup>2</sup>, respectively.

The model further predicts that the  $E_c$  is a linear function of ceramic volume fraction  $\Phi$ . Shown in Fig. 10 is the  $E_c$  vs.  $\Phi$  for 50–65 vol% suspensions including compositions with silica and alumina ceramics. These data can be reasonably represented by straight lines. The slopes, intercepts and correlation coefficients are shown in Table 6. The slope term  $K_6 c_P$  should be numerically equal to (but opposite in sign) to the intercept term  $I_6 c_P$ . The

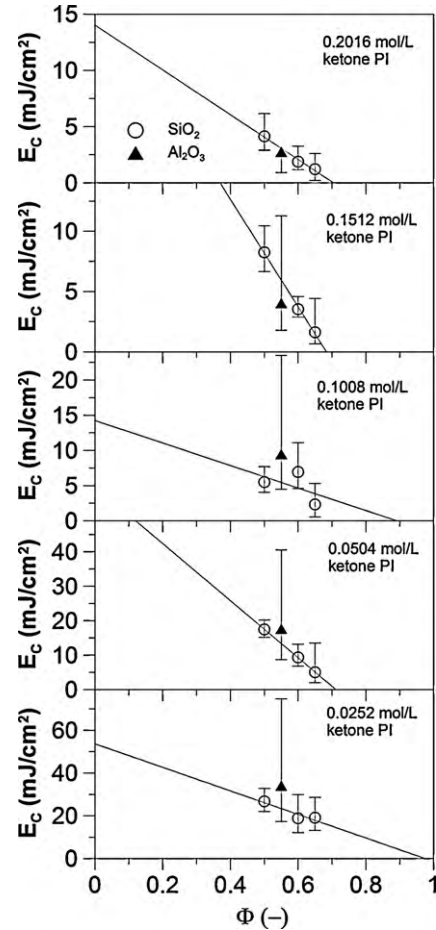


Fig. 10.  $E_c$  vs.  $\Phi$  for  $\text{SiO}_2$  and  $\text{Al}_2\text{O}_3$  suspensions with varying concentration of ketone PI (wrt monomer) and no dye. Cure depth measurements were performed using 355 nm laser

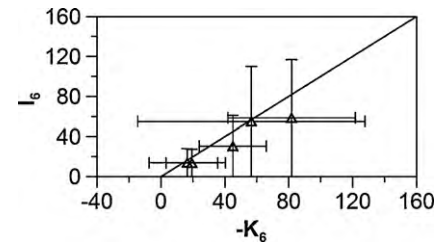


Fig. 11. Intercepts  $I_6$  vs. slopes ( $-K_6$ ) determined from plots  $E_c$  vs.  $\Phi$ . The data appears in Table 6.

slope term and intercept term are approximately similar. Fig. 11 shows the plot of the intercept  $I_6$  vs. the slope  $-K_6$ .

Table 5

Actual slopes, intercepts and correlation coefficients of linear regressions  $E_c$  vs.  $Q_{\text{added}}$  for ~60 vol%  $\text{SiO}_2$  suspensions with ketone PI (0.1008 mol/L wrt monomer). The cure depth measurements were performed using 355 nm laser.

	$r^2$	$K_5$	$I_5$	$\frac{K_5 c_P}{1-\Phi} = \frac{h\nu \gamma Q_{\text{added}}}{\Omega \varepsilon_P} \frac{K_5 c_P}{(1-\Phi)} = \frac{h\nu \gamma Q_{\text{added}}}{\Omega \varepsilon_P}$ (mJ/cm <sup>2</sup> )	$\frac{I_5 c_P}{1-\Phi} = \frac{h\nu}{\Omega} (\gamma Q_{\text{native}} Q_{\text{native}} + \gamma O) \frac{1}{\varepsilon_P}$ (mJ-mol)/(L-cm <sup>2</sup> )
Solid orange HALS	0.98	4100 ± 3000	7 ± 4	1033 ± 756	1.76 ± 1.01
Inhibitor MEHQ	0.87	49 ± 16	10 ± 3	12 ± 4	2.52 ± 0.76
Liquid yellow HALS	0.98	36 ± 25	7 ± 4	9 ± 6	1.76 ± 1.01
Average					2.01





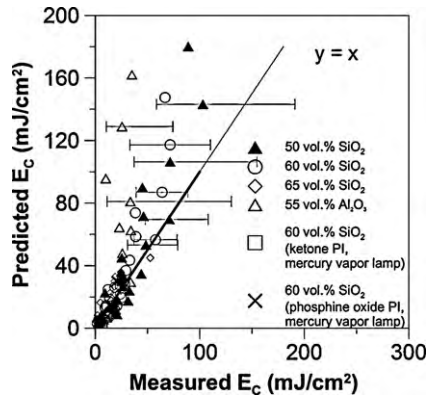


Fig. 12. Predicted  $E_c$  vs. measured  $E_c$  for 50, 60 and 65 vol% SiO<sub>2</sub> suspensions and 55 vol% Al<sub>2</sub>O<sub>3</sub> suspension. The suspensions contained ketone PI and cure depth measurements were performed using the 355 nm laser unless otherwise specified.

of the alumina suspensions. For the silica suspensions most of the predicted  $E_c$  fall within 20% of the experimental  $E_{c,average}$ . Many of the predicted values are larger than the experimental values, but the large uncertainty range of the experimental  $E_c$  values obscures more detailed comparison. The linearized model is able to predict  $E_c$  within about 20% suspensions with  $E_c$  values smaller than 50 mJ/cm<sup>2</sup>. The critical energy for compositions with larger concentrations of dye or inhibitor, where the measured  $E_c$  is larger than 50 mJ/cm<sup>2</sup>, are consistently overestimated by the model. The model assumes that the fraction of photons which are absorbed by the PI to create free radicals is the PI attenuation length divided by the total attenuation length for PI absorption, dye absorption, and scattering, as described by Eq. (20) in the previous paper.<sup>6</sup> Perhaps this is less accurate for dye-dominated or scattering-dominated suspensions, which tend to have higher  $E_c$ . If this approximation under-estimates the fraction of photons absorbed by PIs, the model would overestimate  $E_c$  for these cases.

#### 4. Conclusions

The critical energy dose  $E_c$ , inferred from measurements of cure depth as a function of energy dose, is inversely proportional to the photoinitiator concentration for ketone and phosphine oxide photoinitiators in acrylate monomer with dispersed silica or alumina powders. Inert UV absorbing dyes increase the critical energy dose, with  $E_c$  directly proportional to the dye concentration. The critical energy is directly proportional

to the concentration of quinone inhibitor and hindered amine inhibitors. These results agree with the predictions of the model that presumes that  $E_c$  is the energy dose that produces sufficient free radicals to exhaust native inhibitors. A linearized form of this model predicts linear behavior for  $E_c$  vs. the inverse of the photoinitiator concentration, for  $E_c$  vs. dye concentration, for  $E_c$  vs. inhibitor concentration, and for  $E_c$  vs. ceramic volume fraction. Experiments confirmed these predictions, and provided a self-consistent set of parameters from the slopes and intercepts of the lines.

Values for  $E_c$  predicted from the model using these parameters agreed satisfactorily for  $E_c$  values below about 50 mJ/cm<sup>2</sup>, but overestimate  $E_c$  for suspensions with larger critical energy.

#### Acknowledgements

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